



Fabrication and Process Optimization of Wafer-scale Silicon Nanopillar SERS Substates

Wu, Kaiyu; Schmidt, Michael Stenbæk; Rindzevicius, Tomas; Boisen, Anja

Publication date:
2014

Document Version
Publisher's PDF, also known as Version of record

[Link back to DTU Orbit](#)

Citation (APA):

Wu, K., Schmidt, M. S., Rindzevicius, T., & Boisen, A. (2014). *Fabrication and Process Optimization of Wafer-scale Silicon Nanopillar SERS Substates*. Poster session presented at 40th International Conference on Micro and Nano Engineering, Lausanne, Switzerland.

General rights

Copyright and moral rights for the publications made accessible in the public portal are retained by the authors and/or other copyright owners and it is a condition of accessing publications that users recognise and abide by the legal requirements associated with these rights.

- Users may download and print one copy of any publication from the public portal for the purpose of private study or research.
- You may not further distribute the material or use it for any profit-making activity or commercial gain
- You may freely distribute the URL identifying the publication in the public portal

If you believe that this document breaches copyright please contact us providing details, and we will remove access to the work immediately and investigate your claim.

Introduction

A simple approach for mass-production of wafer-scale Ag capped Si SERS nanopillars is presented and optimized. Recorded SERS spectra exhibit uniform E-field enhancement properties while retaining low background signals over large surface areas ($> \text{cm}^2$). 100 pM of trans-1,2-bis(4-pyridyl)-ethylene (BPE) can be detected.

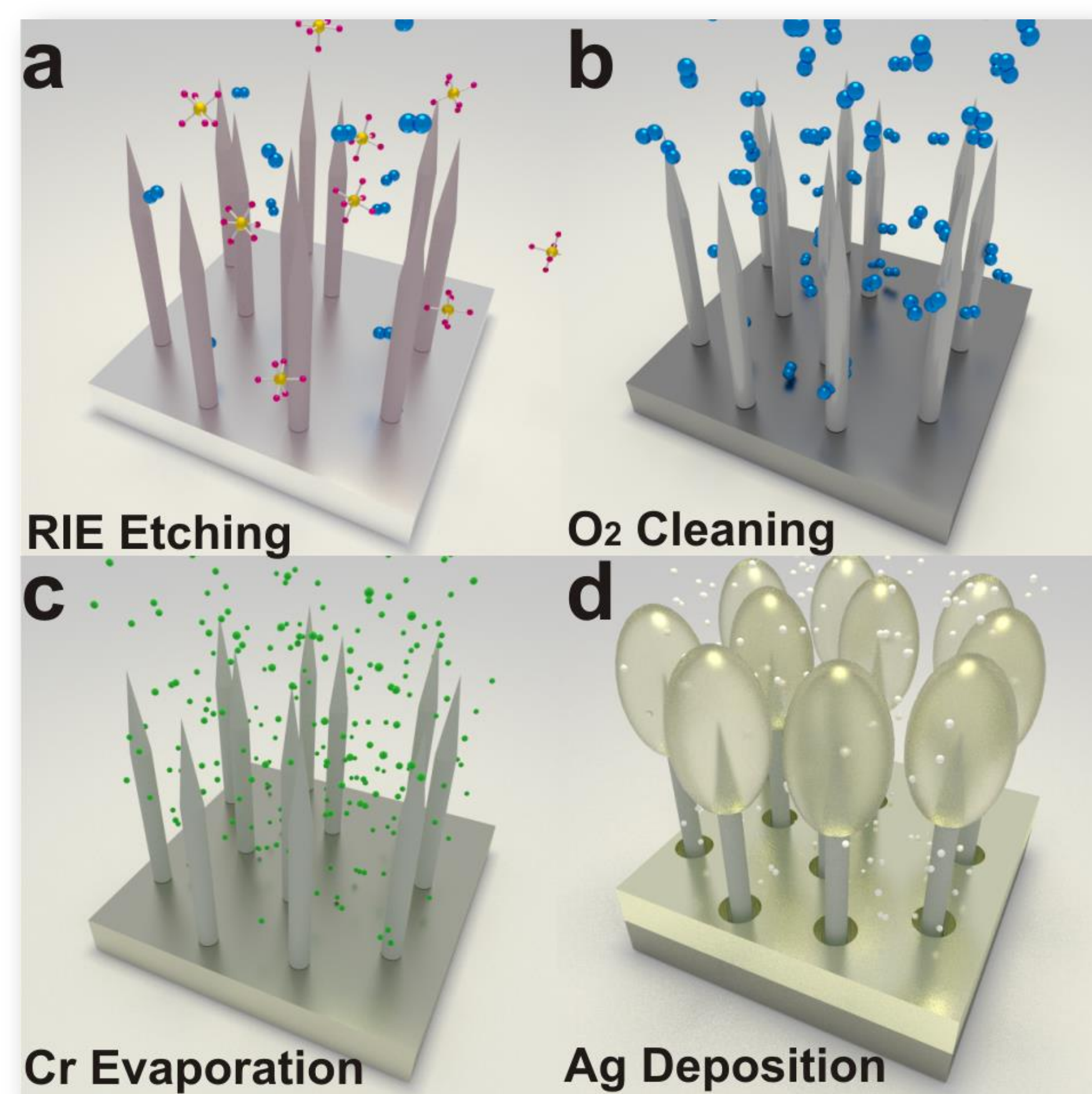


Figure 1. Process steps for Ag NP arrays. (a) Vertically standing Si pillars produced using maskless RIE, $r \approx 20 \pm 3$ nm, $h \approx 300$ – 1200 nm, $\rho_{\text{NP}} \approx 18 \pm 2$ pillars/ μm^2 . (b) Si plasma etching induced surface contaminations are removed using O_2 -plasma, $t = 0$ – 10 min. (c) Deposition of Cr adhesion layer to further reduce SERS background. (d) Evaporation of Ag, $D_{\text{Ag}} = 100$ – 300 nm.

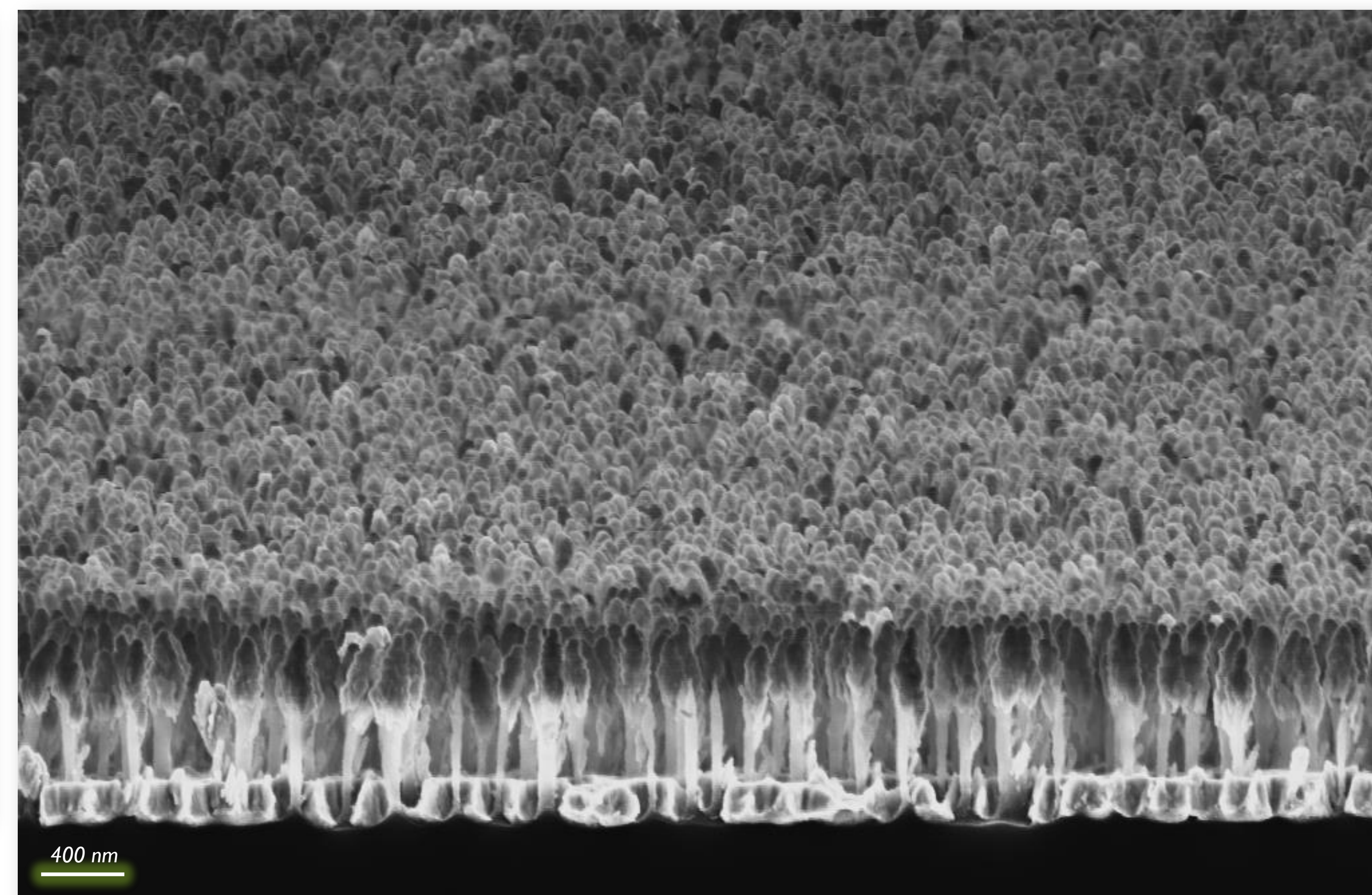


Figure 2. Above: SEM image of the Ag@Si nanopillar surface. Right: (a) Calculated scattering cross section of a Ag@Si nanopillar dimer under different E-field polarization directions. Inset: an SEM image of a Ag@Si pillar dimer. (b) E-field enhancement factor distributions at different excitation wavelengths.

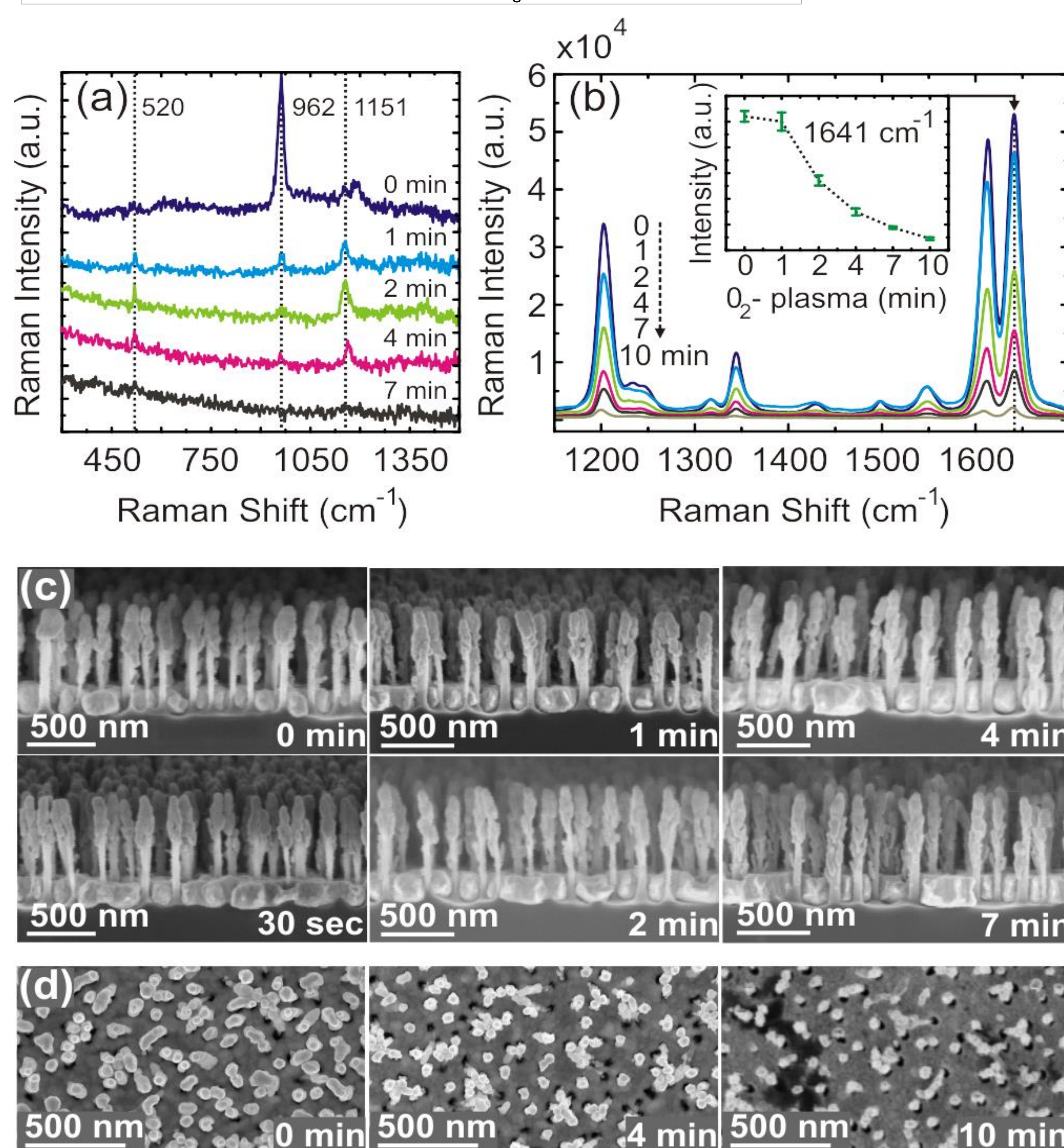
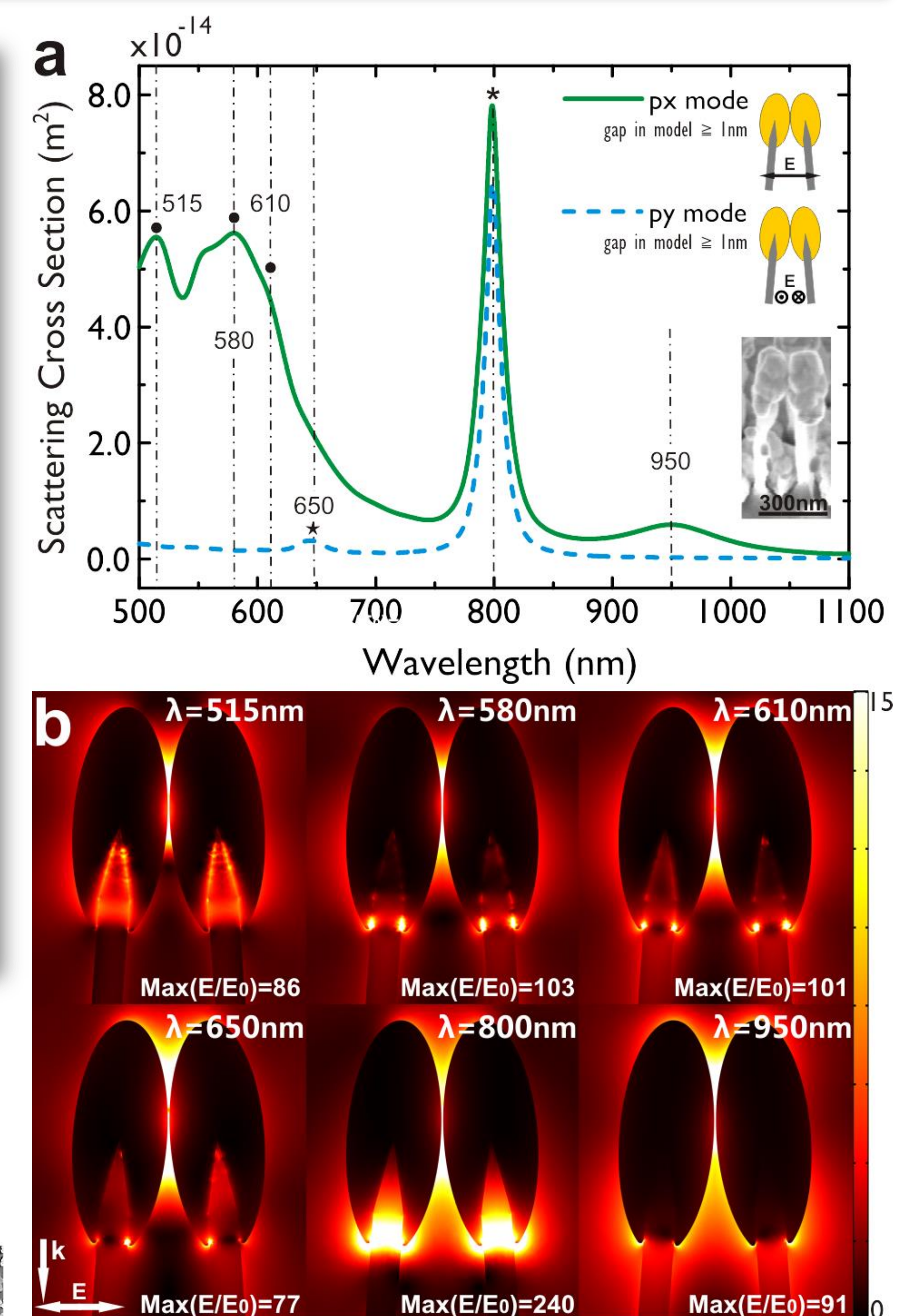


Figure 3. Representative SERS spectra and SEM images of the O_2 -plasma treated Ag NP arrays before, (a) (c), and after, (b) (d), exposure to $1 \mu\text{L}$ of 10 mM BPE in ethanol. Solvent drying pulls Ag NPs together forming nanoclusters of varying size.

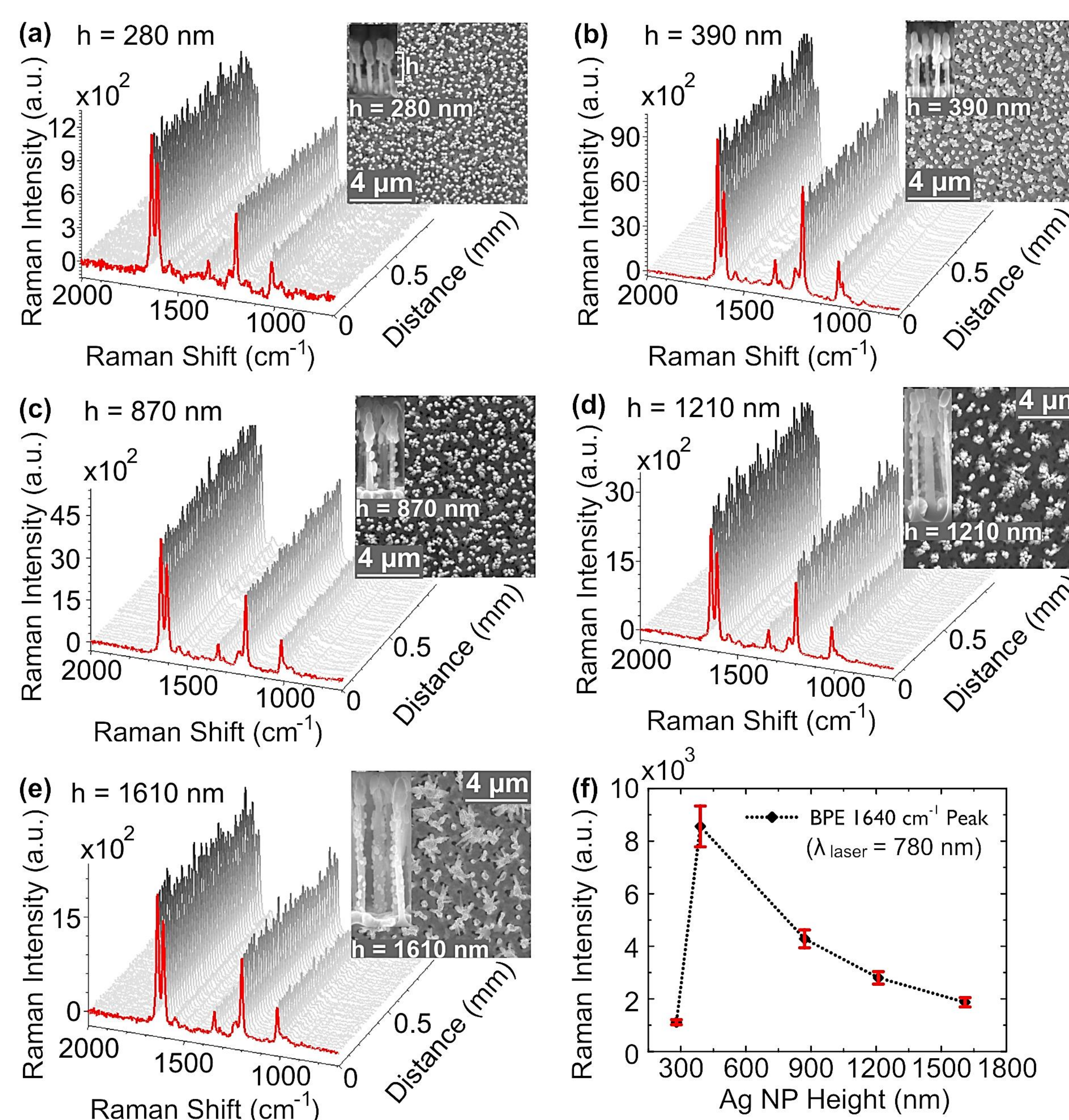


Figure 4. (a) - (e) Line series SERS spectra of 10 mM BPE deposited on Ag@Si NP structures with varying height ($h_p \approx 280$ – 1610 nm). SEM images illustrate the corresponding Ag NP structures (side view) and clustering of Ag NP after exposure to BPE (top view). (f) SERS intensity plots for 1641 cm^{-1} band as a function of NP height averaged by 100 spectra.

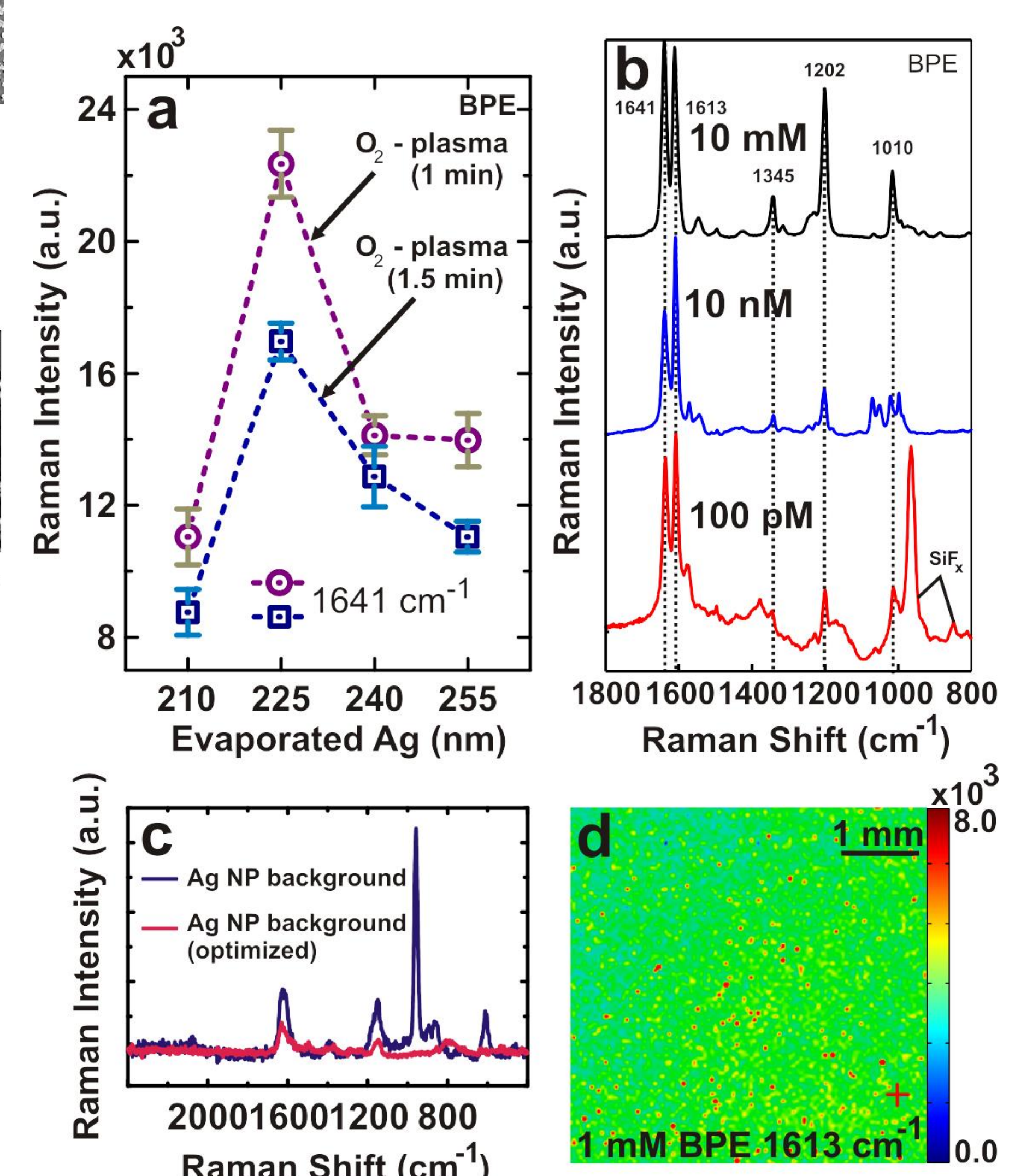


Figure 5: (a) Summary of SERS spectra of 10 mM BPE for substrates with varying Ag metal thickness and O_2 -plasma exposure time. (b) SERS spectra of BPE recorded by optimized NPs that exhibit highest SNR. (c) A comparison between SERS background of standard and optimized Ag NP structures (after leaning). (d) Evaluation of SERS signal uniformity using the optimized substrate.

Discussion

- FEM results in figure 2 show that the most prominent resonance mode is located in the near-infrared spectral region and contributes most to the SERS performance as well as the background of Ag NPs.
- Figure 3 shows that O_2 -plasma exposure reduces the background signal. However process parameters should be carefully chosen to prevent decrease of the EF. Figure 4 shows the optimization of EF by varying the average height of the Ag NP structures.
- Figure 5 shows that a further optimized substrate by varying thickness of Ag evaporated is able to detect 100 pM BPE showing a spectrum which contains five clear Raman vibration modes. The substrate also exhibits high EF uniformity with standard deviations of $\sim 14\%$ across a $5 \text{ mm} \times 5 \text{ mm}$ chip.

Conclusion

A simple approach for mass-production of wafer-scale Ag capped Si SERS nanopillars is presented. Optimization is done to improve SNR of the substrate. Experimental findings suggest that the Ag NP substrates are strong candidates for obtaining a reliable SERS sensing at ultra-low concentrations. The fabrication process is simple, cost-effective, CMOS compatible and could be suitable for mass-production in standard IC foundries utilizing even larger Si carrier wafers.